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Reversed configuration of photocatalyst to exhibit improved properties of basic processes compared to conventional one

Juhong Lian^{1,2}, Yu Qi¹, Yunfeng Bao¹, Zixi Yin³, Yang Zhang¹, Nengcong Yang¹, Naijia Guan², Shengye Jin³, Landong Li^{2*} & Fuxiang Zhang^{1*}

¹State Key Laboratory of Catalysis, iChEM, Dalian Institute of Chemical Physics, Chinese Academy of Sciences, Dalian National Laboratory for Clean Energy, Dalian 116023, China;

²School of Materials Science and Engineering, National Institute for Advanced Materials, Nankai University, Tianjin 300350, China;

³State Key Laboratory of Molecular Reaction Dynamics and Collaborative Innovation Center of Chemistry for Energy Materials (iChEM), Dalian Institute of Chemical Physics, Chinese Academy of Sciences, Dalian 116023, China

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Performances of semiconductor photocatalysts are integrally determined by efficiencies of basic processes such as light absorption, charge separation and surface catalysis, but conventional configurations of photocatalysts normally suffers from the competition of light absorption originating from cocatalyst deposition and limited interface charge separation between the photocatalyst and cocatalyst. Herein we give the first proof-of-concept illustration that a reversed configuration of photocatalysts with a core/shell structure of micro-sized Mo₂N cocatalysts and nano-sized CdS photocatalysts, which exhibits superior solar hydrogen production to the conventional configuration with nano-sized Mo₂N cocatalysts deposited on the surface of CdS photocatalysts. It is revealed that the reversed configuration outperforms the conventional one in all areas of light absorption, charge separation and surface catalysis. Strikingly, the special core/shell structure introduced here can well avoid the competition of light absorption by cocatalysts and make an effective confinement effect to promote the surface catalysis of Mo₂N. Our finding provides an alternative strategy to improve performances of photocatalysts.

photocatalyst, nitride, semiconductor, renewable hydrogen, core/shell

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Hydrogen production using renewable energy has inspired extensive interest since the finding of Fujishima-Honda effect in 1972 [1]. During the past decades, over hundreds of semiconductors have been developed as photocatalysts for promising solar water splitting, but most of them are not active/effective for the surface catalysis to produce hydrogen [2–12]. To address it, noble metals, such as Pt, Rh and Pd, are normally loaded on the surface of semiconductors to act as cocatalysts to promote the surface catalysis [13–17]. Besides the decrease of surface reaction activation barriers/over-

potentials, the cocatalyst is also known to favor the transfer and separation of photogenerated electrons from the semiconductors [18,19]. Accordingly, loading the cocatalysts plays a significant role in promoting the photocatalytic water splitting performances, which has become a general strategy in the field of solar water splitting.

Conventionally, the nanoparticles of noble metal cocatalysts have been loaded on the surface of nano-sized/micro-sized semiconductors. However, this configuration generally confronts some shortcomings. For example, the surface-loaded dark-color noble metals will preferentially absorb light, resulting in decreased light absorption efficiency of the

*Corresponding authors (email: lild@nankai.edu.cn; fxzhang@dicp.ac.cn)

semiconductors [20]. On the other hand, the interface contact between cocatalysts and semiconductors is too insufficient to cause efficient charge separation at the interface, in consideration of the facts that the nanosized semiconductor normally suffers from aggregation [21], while the microsized one owns relatively low surface sites exposed and has a long distance of electron transfer within the bulk [22]. Resulting from the competition of light absorption and inefficient charge transfer, the activities of photocatalysts with the conventional configuration (*i.e.*, loading nanosized cocatalysts on the surface of semiconductors) are mostly unsatisfactory. To address above key issues, it is highly desirable to develop a novel configuration of photocatalysts.

It is additionally noting that the noble metal is too expensive to be for hydrogen production at a large scale, even though the loading of noble metals, especially for metallic platinum, exhibited superior performances [23]. To overcome the dependence on noble metals, the development of non-noble metal-based cocatalysts has recently inspired continuous interest [24]. For example, some transition metal-based nitrides [25-27], carbides [28,29] or sulfides [30,31] have been developed as efficient hydrogen evolution electrocatalysts and cocatalysts of photocatalytic systems for the purpose of replacing the noble metals. Unfortunately, the non-noble metals have been also employed as nanosized cocatalysts for the assembly of conventional configurations of photocatalysts.

Herein we introduce a reversed configuration (compared to the conventional one) of photocatalysts fabricated by depositing nanosized CdS (semiconductor) on the surface of microsized Mo_2N hexagonal rods (cocatalyst). The whole structure can be coarsely described as a core/shell structure. The as-obtained reversed photocatalyst is denoted as $\text{Mo}_2\text{N}@\text{CdS}$. For comparison, the photocatalyst with conventional configurations (called as conventional configuration later) was fabricated by depositing nanosized Mo_2N (cocatalyst) on the surface of bulk CdS (semiconductor), and the corresponding sample is denoted as $\text{Mo}_2\text{N}/\text{CdS}$. Additionally, the nanosized Pt loaded CdS nanoparticles with conventional configurations was also prepared for extended discussion, which is denoted as Pt/CdS. Based on their corresponding optimization, the H_2 evolution rate on the $\text{Mo}_2\text{N}@\text{CdS}$ sample under visible light irradiation is remarkably superior to that of $\text{Mo}_2\text{N}/\text{CdS}$ or Pt/CdS. According to our detailed characterizations and discussion, the superior photocatalytic H_2 evolution rate of the $\text{Mo}_2\text{N}@\text{CdS}$ sample is mainly attributed to its special advantages in all the areas of light absorption, charge separation and surface catalysis: 1) the core/shell structure in the reversed configuration can ensure the preferential light absorption of the CdS semiconductor to even avoid the competition of light harvesting caused by the Mo_2N cocatalyst; 2) much better charge separation is observed for the reversed configuration

with respect to the conventional ones; 3) the core/shell structure makes an effective confinement effect to promote the surface catalysis of Mo_2N .

As seen in the illustration of Figure 1(a), the $\text{Mo}_2\text{N}@\text{CdS}$ photocatalyst was basically constructed by three steps. First of all, the MoO_3 hexagonal rods were prepared by an oil bath method. Secondly, the MoO_3 hexagonal rods were employed as precursors to prepare Mo_2N hexagonal rods under the flow of ammonia. Finally, the microsized Mo_2N hexagonal rods were employed as cocatalysts, and on the surface a calculated amount of CdS was deposited by a simple hydrothermal process to obtain the reversed core/shell configuration. The detailed experimental procedure is illustrated in the Experimental section. If not specially stated, the samples denoted as $\text{Mo}_2\text{N}@\text{CdS}$, $\text{Mo}_2\text{N}/\text{CdS}$ and Pt/CdS stand for those with optimized content of cocatalysts, respectively.

As given in Figure 1(b), a single phase of MoO_3 and Mo_2N can be confirmed according to the observation that all the X-ray diffraction peaks are well indexed to the hexagonal phase of MoO_3 (PDF# 21-0569) and the cubic phase of Mo_2N (PDF# 25-1366), respectively. After depositing CdS on the surface of Mo_2N , the diffraction peaks attributed to the Mo_2N phase are well kept, accompanying with the diffraction peaks of hexagonal phases of CdS (PDF# 41-1049). This demonstrates the good composite of both Mo_2N and CdS by the experimental procedure. The hexagonal MoO_3 and Mo_2N rods can be separately observed in Figure 1(c, d). Intriguingly, the Mo_2N rods with the length of 5–15 μm are well dispersed free of aggregation. Additionally, the CdS nano-

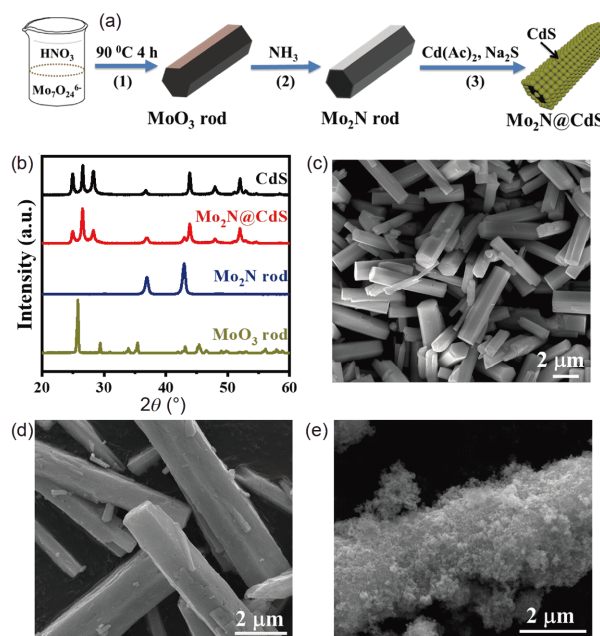


Figure 1 (a) An illustration on the preparation of $\text{Mo}_2\text{N}@\text{CdS}$ core-shell structures. (b) XRD patterns of the MoO_3 hexagonal rods, Mo_2N hexagonal rods, $\text{Mo}_2\text{N}@\text{CdS}$ and CdS samples. SEM images of the MoO_3 hexagonal rods (c), Mo_2N hexagonal rods (d) and $\text{Mo}_2\text{N}@\text{CdS}$ core-shell structures (e) (color online).

particles with the particle size of *ca.* 30–50 nm are mainly dispersed around the Mo₂N hexagonal rods (Figure 1(e)) to form the reversed core/shell structure. High resolution scanning electron microscopy (HRSEM, see Figure S2, Supporting Information online) images more clearly show that CdS nanoparticles grow on the surface of Mo₂N rods uniformly. The SEM image of pristine CdS in Figure S1 shows an aggregated morphology with the particle sizes of 30–50 nm. The typical structures and morphologies of the Pt/CdS and Mo₂N/CdS samples are given in Figures S3–S5, respectively. As expected, the diffraction peaks assigned to the deposited Pt or Mo₂N nanoparticles are not observed, demonstrating their small size and/or amorphous structures. However, obvious aggregation of the Mo₂N nanoparticles on the surface of bulk CdS can be observed (Figure S4). The Pt nanoparticles by the photodeposition method can be also observed on the surface of CdS (Figure S5).

Photocatalytic H₂ evolution rates were evaluated in the aqueous solution containing lactic acid (as hole scavengers) under visible light irradiation. All the samples show no obvious H₂ evolution under dark condition, and no obvious H₂ was detected for the pure Mo₂N hexagonal rod under visible light, either. The CdS shows a low H₂ evolution rate (*ca.* 45 μmol/h). This means that CdS can act as photocatalysts for the H₂ evolution, and the Mo₂N can only work as cocatalysts. The dependence of content of cocatalysts on the initial H₂ evolution rate was first optimized (Figure S6). Figure 2(a) gives their typical time courses of H₂ evolution on the optimized samples, in which the amount of generated H₂ almost linearly increased after 1 h induced irradiation, demonstrating their good photochemical stability in the experimental region. Figure 2(b) further compares their optimal average activities, based on which the Mo₂N@CdS sample exhibits remarkably superior performance with respect to the Pt/CdS or Mo₂N/CdS samples. To evaluate the stability of reversed Mo₂N@CdS core/shell structures, the XRD patterns of Mo₂N@CdS were measured after the photocatalytic H₂ evolution reaction. As shown in Figure S7, no obvious changes of XRD patterns were observed before and after the photocatalytic reaction, demonstrating that the Mo₂N@CdS core/shell structure is stable under the experimental conditions.

In order to get insight into the superior performance of the reversed configuration with respect to the conventional ones, the typical three samples are thus analyzed and compared in detail. Firstly, their charge separation performances were analyzed and compared by measuring their steady-state photoluminescence (PL), time-resolved photoluminescence (TRPL) spectra, transient photocurrent response and electrochemical impedance spectra (EIS). As demonstrated by the PL results in Figure 3(a), all the samples show the fluorescence signals at 525 nm due to intrinsic band-band transitions while the much lower PL intensity of Mo₂-

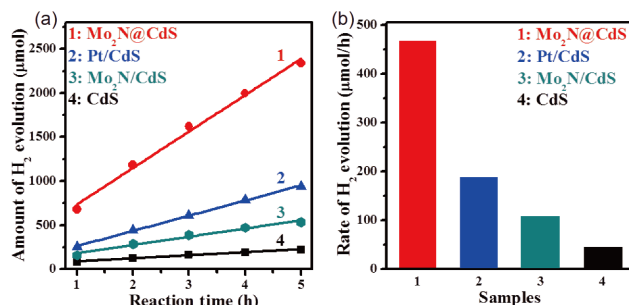


Figure 2 Comparison of the photocatalytic H₂ evolution performance on typical samples. (a) Time courses of H₂ evolution; (b) the average H₂ generation rate. Reaction conditions: 20 mg CdS; 100 mL, 10 vol% lactic acid aqueous solution; 300 W xenon lamp ($\lambda > 420$ nm); reaction time: 5 h (color online).

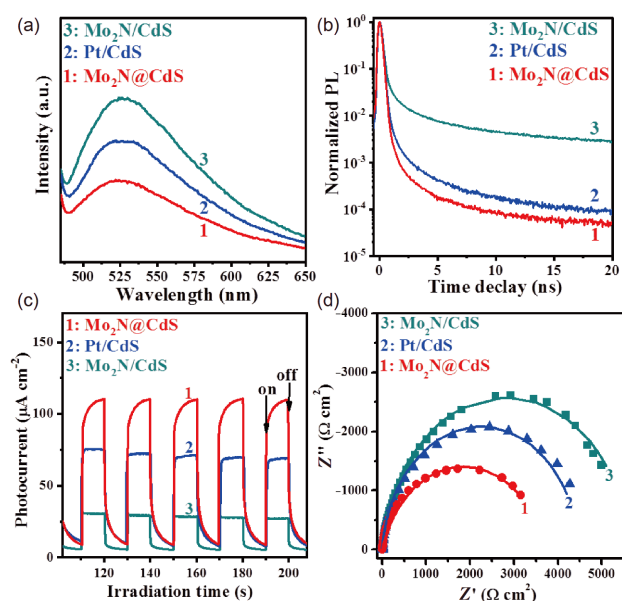


Figure 3 The charge separation performance of the typical three samples. (a) Steady PL spectra; (b) TRPL decay spectra; (c) transient photocurrent responses recorded at the potential of 0.9 V vs. RHE in 0.5 M Na₂S and 0.5 M Na₂SO₃; (d) Nyquist plots tested at 0.9 V vs. RHE; frequency: 0.1 Hz–10 kHz (color online).

N@CdS than the other two samples is observed, indicating that the reversed Mo₂N@CdS shows more effective charge separation between CdS and Mo₂N cocatalysts than the other two samples. TRPL spectra were further performed to measure the charge separation dynamics. The cocatalysts can accept the photogenerated electrons from excited photocatalysts to allow additional nonradiative decay channels, and decrease the fluorescence lifetime [32–36]. As shown in Figure 3(b), Mo₂N@CdS displays the fastest PL decay lifetime, which implies that the efficient contact interface of reserved Mo₂N@CdS configurations can accelerate the electrons transfer from CdS to Mo₂N compared to the conventional configuration samples. The superior charge separation between photocatalysts and cocatalysts will lead to excellent photoelectrocatalytic performances. Expectedly,

transient photocurrent results reveal that reserved Mo₂N@CdS exhibits enhanced photocurrent responses compared with Pt/CdS or Mo₂N/CdS samples (Figure 3(c)), suggesting that this reserved Mo₂N@CdS can promote the separation and transfer of photogenerated electron-hole pairs. The current enhancement and decay during the light-on and -off period may result from the charge and discharge of the sample [37]. Meanwhile, the Mo₂N@CdS shows much smaller electrolyte interface charge transfer resistance (R_{ct}) than that of Pt/CdS or Mo₂N/CdS, further demonstrating better charge separation and transfer of the Mo₂N@CdS (Figure 3(d)). It is worth noting that trends of the intensity of PL spectra, TRPL decay lifetime, transient photocurrent responses and the charge transfer resistance on the three typical samples are all in good accordance with their activity order. This demonstrates that the charge separation plays a remarkable role in determining their activities.

Encouraged by the superior charge separation performance on the reversed configuration, we continuously investigated the feature of surface catalysis and light harvesting in the Mo₂N@CdS sample. As seen in Figure 4(a), the Mo₂N rods show the effective electrocatalytic water reduction, confirming its function of cocatalysts for the H₂ evolution. Intriguingly, the electrocatalytic performance of Mo₂N rods is remarkably promoted after the deposition of CdS which is not active for the water reduction at all. This means that a nano-confined effect may result from the special core/shell structure. The nano-confined effect can be explained by electron interactions and stack porous structure of CdS nanoparticles. Firstly, the XPS result in Figure 4(b) shows that the binding energy of Mo–N has an obvious negative shift (228.0 eV) after the deposition of CdS nanolayers compared with the pristine Mo₂N rods (228.7 eV), suggesting a strong electron interaction between Mo₂N and CdS in the reversed configuration. The strong electron interactions play an important role in promoting the electrochemical hydrogen production, which benefits to the activation of H₂O [38]. The survey spectrum shows the existence of Cd, S, Mo, N and O

elements in Figure S8. In addition, the porous structure originating from the stack of CdS nanoparticles can be formed, in which the H₂O molecules could stay longer and have more opportunities to collide with the Mo₂N surface. The obvious electron exchange and the porous structures are the basic features of nano-confined effect [39–43]. This is consistent with previous reports which the pore-confined space makes a significant promotion effect on the electrochemical efficiency [44,45]. Additionally, the reversed configuration of Mo₂N@CdS shows weaker characteristic absorption of Mo₂N than conventional configurations of Mo₂N/CdS (Figure S9). In order to further analyze the thickness of CdS on the light absorption, CdS films with different thickness were prepared and characterized. As given in Figure 4(c), the transmittance of solar light is strongly dependent on the thickness of CdS layers, and almost complete light absorption before absorption edge can be obtained when the thickness of CdS layers is beyond 350 nm. This means that the competition of light absorption originating from the Mo₂N cocatalyst will be completely avoided once the thickness of CdS shell photocatalysts is over 350 nm.

It has been widely known that the activity of one photocatalyst is integrally determined by the efficiencies of light absorption, charge separation and surface catalysis [46]. On the basis of our above analysis and discussion, the photocatalyst with reversed configurations (*i.e.*, Mo₂N@CdS) is not only favorable for the charge separation but also for the surface catalysis due to the unexpected nano-confined effect. Additionally, the reversed configuration can effectively avoid the competition of light absorption resulting from the cocatalyst, while this is impossible for the conventional configuration. To further illustrate the influence of light absorption competition, we compared a series of reversed photocatalysts with different thicknesses of CdS by adjusting the mass ratio of CdS to Mo₂N rods. As given in Figure S6 (a), the photocatalytic activities of Mo₂N@CdS undergo first increase and subsequent decrease, revealing that the photocatalytic activity is really related to the light absorption ef-

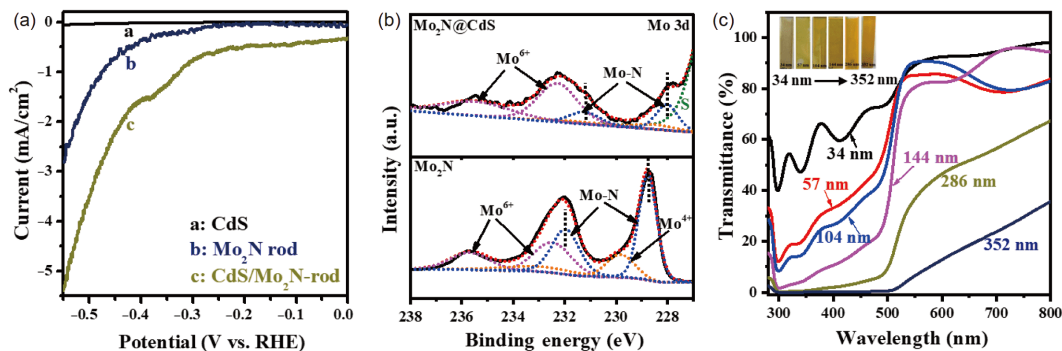


Figure 4 The performances of surface catalysis and light absorption of the Mo₂N@CdS. (a) Current density curves as a function of the applied potential of the CdS, Mo₂N rod and CdS/Mo₂N-rod samples; (b) high-resolution XPS spectra of Mo 3d for Mo₂N rods and Mo₂N@CdS samples; (c) UV-Vis transmittance spectra of CdS films on FTO (color online).

iciency, and the thickness of the CdS photocatalyst with saturated light absorption should be favorable. Additionally, it is reasonable to deduce that the increasing thickness of CdS layers (*i.e.*, increasing the mass ratio of CdS to Mo₂N) not only favors the light absorption but also simultaneously causes much longer charge transfer distance to render the charge separation more difficult. As an integral effect of light absorption and charge separation, the optimal mass ratio of CdS to Mo₂N is thus observed. In this work, the CdS thickness on the sample with the optimal activity was examined by TEM images (Figure S10), and the average thickness of CdS shells is about 370 nm that is beyond the thickness with the complete light absorption before the absorption of CdS films (350 nm, Figure 4(c)). In this case, the competition of light absorption caused by the Mo₂N cocatalysts is well avoided.

Summarily, we designed and realized an unconventional reversed configuration of photocatalysts using microsized Mo₂N (noble metal free) rods as cocatalyst cores and nano-sized CdS as model semiconductor photocatalyst shells, which first exhibit superior solar hydrogen production performances with respect to conventional configurations. The reversed core/shell structure not only favors the charge separation, but also efficiently avoids the competition of light harvesting caused by the loaded cocatalysts and makes an unexpected confinement effect for the better catalysis of water reduction. Integrated by the synergistic promotion of light absorption, charge separation and surface catalysis, the photocatalytic performance of the reversed configuration is obviously superior to that of the conventional ones, which is even higher than that of the noble Pt loaded CdS with conventional configurations. This work gives a proof-of-concept illustration on the potential advantages of reversed configurations introduced here, which may be extended into the fabrication of other reversed nanocomposites with inner 1D (or 2D) cocatalysts and outer 0D photocatalysts to promote solar-to-chemical conversion.

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Conflict of interest The authors declare that they have no conflict of interest.

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- Li Q, Guo B, Yu J, Ran J, Zhang B, Yan H, Gong JR. *J Am Chem Soc*, 2011, 133: 10878–10884
- Hitoki G, Ishikawa A, Takata T, Kondo JN, Hara M, Domen K. *Chem Lett*, 2002, 31: 736–737
- Chen S, Shen S, Liu G, Qi Y, Zhang F, Li C. *Angew Chem Int Ed*, 2015, 54: 3047–3051
- Chen S, Qi Y, Ding Q, Li Z, Cui J, Zhang F, Li C. *J Catal*, 2016, 339: 77–83
- Zheng D, Cao XN, Wang X. *Angew Chem Int Ed*, 2016, 55: 11512–11516
- Hitoki G, Takata T, Kondo JN, Hara M, Kobayashi H, Domen K. *Chem Commun*, 2002, 1698–1699
- Maeda K, Terashima H, Kase K, Higashi M, Tabata M, Domen K. *BCSJ*, 2008, 81: 927–937
- Kailasam K, Schmidt J, Bildirir H, Zhang G, Blechert S, Wang X, Thomas A. *Macromol Rapid Commun*, 2013, 34: 1008–1013
- Kiss B, Didier C, Johnson T, Manning TD, Dyer MS, Cowan AJ, Claridge JB, Darwent JR, Rosseinsky MJ. *Angew Chem Int Ed*, 2014, 53: 14480–14484
- Liu GL, Chen CC, Ji HW, Ma WH, Zhao JC. *Sci China Chem*, 2012, 55: 1953–1958
- Ling X, Xu Y, Wu S, Liu M, Yang P, Qiu C, Zhang G, Zhou H, Su C. *Sci China Chem*, 2020, 63: 386–392
- Schweinberger FF, Berr MJ, Döblinger M, Wolff C, Sanwald KE, Crampton AS, Ridge CJ, Jäckel F, Feldmann J, Tschurl M, Heiz U. *J Am Chem Soc*, 2013, 135: 13262–13265
- Dukovic G, Merkle MG, Nelson JH, Hughes SM, Alivisatos AP. *Adv Mater*, 2008, 20: 4306–4311
- Li Z, Zhang F, Han J, Zhu J, Li M, Zhang B, Fan W, Lu J, Li C. *Catal Lett*, 2018, 148: 933–939
- Maeda K, Teramura K, Domen K. *J Catal*, 2008, 254: 198–204
- Wen B, Ma JH, Chen CC, Ma WH, Zhu HY, Zhao JC. *Sci China Chem*, 2011, 54: 887–897
- Wang Q, Hisatomi T, Jia Q, Tokudome H, Zhong M, Wang C, Pan Z, Takata T, Nakabayashi M, Shibata N, Li Y, Sharp ID, Kudo A, Yamada T, Domen K. *Nat Mater*, 2016, 15: 611–615
- Yang J, Wang D, Han H, Li C. *Acc Chem Res*, 2013, 46: 1900–1909
- Ran J, Gao G, Li FT, Ma TY, Du A, Qiao SZ. *Nat Commun*, 2017, 8: 13907
- Wang D, Li X, Zheng LL, Qin LM, Li S, Ye P, Li Y, Zou JP. *Nanoscale*, 2018, 10: 19509–19516
- Xiao M, Luo B, Lyu M, Wang S, Wang L. *Adv Energy Mater*, 2018, 8: 1701605
- Yan H, Yang J, Ma G, Wu G, Zong X, Lei Z, Shi J, Li C. *J Catal*, 2009, 266: 165–168
- Shi Y, Zhang B. *Chem Soc Rev*, 2016, 45: 1529–1541
- Shi X, Wu A, Yan H, Zhang L, Tian C, Wang L, Fu H. *J Mater Chem A*, 2018, 6: 20100–20109
- Yan H, Xie Y, Jiao Y, Wu A, Tian C, Zhang X, Wang L, Fu H. *Adv Mater*, 2018, 30: 1704156
- Zhu Y, Chen G, Xu X, Yang G, Liu M, Shao Z. *ACS Catal*, 2017, 7: 3540–3547
- Chen YY, Zhang Y, Jiang WJ, Zhang X, Dai Z, Wan LJ, Hu JS. *ACS Nano*, 2016, 10: 8851–8860
- Wang S, Wang J, Zhu M, Bao X, Xiao B, Su D, Li H, Wang Y. *J Am Chem Soc*, 2015, 137: 15753–15759
- Li Y, Wang H, Xie L, Liang Y, Hong G, Dai H. *J Am Chem Soc*, 2011, 133: 7296–7299
- Jaramillo TF, Jørgensen KP, Bonde J, Nielsen JH, Hørcher S, Chorkendorff I. *Science*, 2007, 317: 100–102
- Tu W, Li Y, Kuai L, Zhou Y, Xu Q, Li H, Wang X, Xiao M, Zou Z. *Nanoscale*, 2017, 9: 9065–9070
- Wang W, Zhu S, Cao Y, Tao Y, Li X, Pan D, Phillips DL, Zhang D, Chen M, Li G, Li H. *Adv Funct Mater*, 2019, 29: 1901958
- Shi H, Long S, Hu S, Hou J, Ni W, Song C, Li K, Gurzadyan GG, Guo X. *Appl Catal B-Environ*, 2019, 245: 760–769

- 35 Chen H, Sun Z, Ye S, Lu D, Du P. *J Mater Chem A*, 2015, 3: 15729–15737
- 36 Zhang H, Lin J, Li Z, Li T, Jia X, Wu XL, Hu S, Lin H, Chen J, Zhu J. *Catal Sci Technol*, 2019, 9: 502–508
- 37 Liu G, Shi J, Zhang F, Chen Z, Han J, Ding C, Chen S, Wang Z, Han H, Li C. *Angew Chem Int Ed*, 2014, 53: 7295–7299
- 38 Wang B, Huang H, Huang M, Yan P, Isimjan TT, Yang X. *Sci China Chem*, 2020, <https://doi.org/10.1007/s11426-019-9721-0>
- 39 Fu Q, Bao X. *Nat Catal*, 2019, 2: 834–836
- 40 Wang T, Gao L, Hou J, Herou SJA, Griffiths JT, Li W, Dong J, Gao S, Titirici MM, Kumar RV, Cheetham AK, Bao X, Fu Q, Smoukov SK. *Nat Commun*, 2019, 10: 1340–1348
- 41 Wang Y, Mao J, Meng X, Yu L, Deng D, Bao X. *Chem Rev*, 2019, 119: 1806–1854
- 42 Cui T, Dong J, Pan X, Yu T, Fu Q, Bao X. *J Energy Chem*, 2019, 28: 123–127
- 43 Li H, Guo C, Fu Q, Xiao J. *J Phys Chem Lett*, 2019, 10: 533–539
- 44 Bae JH, Han JH, Chung TD. *Phys Chem Chem Phys*, 2012, 14: 448–463
- 45 Zhang J, Li CM. *Chem Soc Rev*, 2012, 41: 7016–7031
- 46 Moniz SJA, Shevlin SA, Martin DJ, Guo ZX, Tang J. *Energy Environ Sci*, 2015, 8: 731–759